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10/524620

PCT/JP03/10673 (Our File No. KRC-76)

DT05 Rec'd PCT/PTO 15 FEB 2005

## CONTINUOUS CASTING OF MOLTEN STEEL FOR SHEET METAL

### TECHNICAL FIELD

The present invention relates to continuous casting of molten steel to be formed as a sheet metal, and more particularly to a refractory material for use in the continuous casting.

### BACKGROUND ART

In late years, the need for strict quality control of steel products has become increasingly prominent. Under these circumstances, in the continuous casting of aluminum-deoxidized steel (hereinafter referred to as "aluminum-killed steel") to be formed as high-quality steel, such as sheet metal, a good deal of effort has been made to prevent alumina from being attached onto a casting nozzle for pouring molten steel from a tundish into a mold.

The alumina attached onto the casting nozzle will be combinedly accumulated, and formed as a large-size impurity. Then, the alumina will be mixed in molten steel flow, and incorporated into slabs as a large-size inclusion to cause defect in slabs or deterioration in quality thereof.

As one of measures against this problem, there has been known a method comprising injecting argon gas from the inner surface of a casting nozzle toward molten steel to physically prevent the attachment of alumina. However, if an excessive amount of argon gas is injected, resulting gas bubbles will be incorporated into slabs to cause the formation of pinholes or defect. Thus, this method is not exactly a sufficient measure due to the restriction in the allowable injection amount of argon gas.

There has also been known a method intended to provide an anti-alumina-buildup function to a refractory material itself. This method is directed to prevent the buildup of alumina by preparing a refractory brick containing CaO and causing the reaction between CaO and alumina attached to the brick to form a low-melting-point compound. For example, Japanese Patent Laid-Open Publication No. 11-506393 discloses a casting nozzle using a refractory material comprising in combination graphite, and dolomite clinker having a primary component

consisting of CaO and MgO.

In order to obtain the anti-alumina-buildup function, this material can be applied to the surface of an inner hole of a submerged nozzle for casting of aluminum-killed steel. In this case, while the amount of alumina to be attached on the surface of the inner hole of the submerged nozzle is actually reduced, a large-size inclusion is often detected from slabs to be formed as a sheet metal. The inclusion causes occurrence of flaws in a rolling process of the slabs. In particular, the inclusion has a serious impact on slabs to be formed as a thin sheet metal.

## DISCLOSURE OF INVENTION

It is therefore an object of the present invention to provide a refractory material for use in a continuous casting nozzle, which contains CaO · MgO-based clinker having CaO as a mineral phase having an anti-alumina-buildup effect, so as to allow the amount of a large-size inclusion in slabs during casting of aluminum-killed steel to be drastically reduced.

Through various researches on inclusions detected from slabs in case of using a continuous casting nozzle made of a refractory material containing CaO · MgO-based clinker having an anti-alumina-buildup effect, the inventors found that large-size inclusions having a diameter of 50  $\mu\text{m}$  or more comprise magnesia as a primary component. The inventors assumed that the magnesia as the inclusions was originated from the refractory material containing CaO · MgO-based clinker.

FIG. 1 shows the distribution state of MgO in CaO · MgO-based clinker by an electron micrograph. As seen in this electron micrograph, MgO grains are independently dispersed over the CaO · MgO-based clinker in the form of small grains, because no compound is formed between CaO and MgO.

And FIG. 2 shows the relationship between the mean diameter of MgO particles in the CaO · MgO-based clinker and the size of MgO inclusion in a cast aluminum-killed slab obtained by using CaO · MgO-based clinker as shown in the electron micrograph of FIG. 1. As seen in FIG. 2, it is proved that there is a positive correlation between the size of the MgO grain particles in the clinker and the size of the inclusions, and the size of the MgO grain particles is

analogous to the size of the inclusions.

When a refractory material containing  $\text{CaO} \cdot \text{MgO}$ -based clinker is used in a refractory component of a continuous casting apparatus for casting aluminum-killed steel, at the surface of the refractory to be in contact with molten steel, alumina dispersed over the steel reacts with  $\text{CaO}$  in the  $\text{CaO} \cdot \text{MgO}$ -based clinker to form an  $\text{Al}_2\text{O}_3 \cdot \text{CaO}$ -based low-melting-point compound, and the low-melting-point compound is then spilled off from the surface of the refractory component with the flow of the molten steel.

The  $\text{Al}_2\text{O}_3 \cdot \text{CaO}$ -based compound released from the surface of the refractory component is readily dispersed over the molten steel, and less apt to be formed as a large-size inclusion. Thus, an adverse affect on the quality of slabs can be minimized.

Even if the  $\text{Al}_2\text{O}_3 \cdot \text{CaO}$ -based compound has a large size, it will be relatively harmless because it is relatively soft, and can be thinly drawn out in a rolling process.

In contrast,  $\text{MgO}$  in the clinker is apt to flow in the molten steel while maintaining its original grain size, due to its low reactivity as compared to  $\text{CaO}$ . Further,  $\text{MgO}$  has a high melting point and high hardness. If a large-size  $\text{MgO}$  grain is mixed in slabs, it will cause flaws in the rolling process or deterioration in quality of the slabs. In addition, it is often the case that the diameter of  $\text{MgO}$  crystal grains in the clinker is equal to that of  $\text{MgO}$ -based inclusions in the slabs as shown in FIG. 2, because  $\text{MgO}$  in the clinker is apt to flow in the molten steel while maintaining its original grain size. Thus, in view of reducing the amount of large-size inclusions in slabs, it is essential to miniaturize  $\text{MgO}$  crystal grains in the  $\text{CaO} \cdot \text{MgO}$ -based clinker.

Based on the above knowledge, in order to solve the problem of the  $\text{MgO}$ -based inclusions caused by the  $\text{CaO} \cdot \text{MgO}$ -based clinker-containing refractory material for use in a continuous casting apparatus for molten steel to be formed as a sheet metal, the present invention provides a refractory material containing 20 mass% or more of  $\text{CaO} \cdot \text{MgO}$ -based clinker. In the refractory material, 60% or more of  $\text{MgO}$  grains have a diameter of 50  $\mu\text{m}$  or less. The refractory material is used at least at a portion of a continuous casting apparatus to be in contact with molten steel.

Generally, in view of sheet metal products, it is desired to minimize inclusions having a

diameter of 50  $\mu\text{m}$  or more, and to reduce the diameter of MgO grains in CaO · MgO-based clinker as much as possible. In a practical sense, if 60% or more of MgO grains in the clinker have a diameter of 50  $\mu\text{m}$  or less, the casting of aluminum-killed steel for sheet metal can be performed without any problem. Thus, the CaO · MgO-based clinker is prepared to allow 60% or more of MgO grains therein to have a diameter of 50  $\mu\text{m}$  or less. In particular, steel for beverage tin cans is required to be free of inclusion having a diameter of 50  $\mu\text{m}$  or more. Thus, in a casting process of molten steel to be formed as beverage tin cans, it is desired to use clinker containing MgO crystal grains having a smaller mean diameter, for example, of 20  $\mu\text{m}$  or less. The diameter of MgO grain in the CaO · MgO-based clinker herein is determined by separating MgO crystal grains and CaO grains in an electron micrograph of the clinker from each other using an image analysis apparatus, converting the area of each of the MgO grains into a circular area, and measuring the diameter of the circular area.

According to preparation methods, CaO · MgO-based clinker is classified into three types: synthetic dolomite clinker, natural dolomite clinker and electro-fused CaO · MgO-based clinker. The synthetic dolomite clinker is prepared by burning mixed particles of Ca (OH)<sub>2</sub> and Mg (OH)<sub>2</sub> at a high temperature. The natural dolomite clinker is prepared by burning dolomite as a natural product at a high temperature. The electro-fused CaO · MgO-based clinker is prepared by arc-melting a material containing CaO components and MgO components, and cooling/solidifying the arc-melted material. The diameter of MgO crystal grains in the synthetic CaO · MgO-based clinker can be changed by controlling the particle size of a starting material. Specifically, Mg (OH)<sub>2</sub> particles in the starting material can be arranged to have a small particle size and enhanced dispersibility to reduce the diameter of MgO crystal grains in the clinker.

The diameter of MgO grains in the natural dolomite clinker is varied depending on native dolomite minerals. Thus, the natural dolomite clinker can be prepared by using one native dolomite mineral providing a desired diameter of MgO grains. The diameter of MgO crystal grains in the electro-fused CaO · MgO-based clinker can be controlled by adjusting the cooling rate in the solidification process.

In dolomite clinker prepared by using natural dolomite, the chemical components of CaO

and MgO have an approximately constant mass ratio of about 60 : 40 (CaO : MgO). By contrast, in the synthetic dolomite clinker and the electro-fused CaO · MgO-based clinker, the mass ratio of CaO : MgO can be freely changed. In this case, if the MgO component is excessively increased, MgO crystal grains will be combined with each other to form an undesirably enlarged MgO crystal grain. Thus, the content of MgO is preferably set to be less than 50 mass%.

The refractory material containing 20 mass% or more of CaO · MgO-based clinker may be prepared by adding an organic binder to CaO · MgO-based clinker, uniformly kneading the compound to form a green body, and burning the green body at about 1600°C, or by adding phenol resin and 10 to 40% of graphite to CaO · MgO-based clinker, uniformly kneading the compound to form a green body, and burning the green body at about 1000°C in a reduction atmosphere. Through this process, the refractory material can have an anti-alumina-buildup function.

For example, the refractory material without containing graphite is suitable for use in a refractory component of a continuous casting apparatus which is applied with relatively low thermal shock in use, such as an upper nozzle, a sliding nozzle, a lower nozzle or a stopper head. The refractory material containing graphite is suitable for use in a refractory component of a continuous casting apparatus which is applied with relatively high thermal shock in use, such as a submerged nozzle, a long nozzle, or a long stopper. However, the use of such refractory materials is not limited to the above examples. What is important is to appropriately adjust the amount of graphite depending on use conditions.

In view of the anti-alumina-buildup effect, in addition to 20 mass% or more of CaO · MgO-based clinker including CaO, the refractory material of the present invention may contain carbon or another clinker, such as CaO clinker, ZrO<sub>2</sub> clinker, ZrO<sub>2</sub> · CaO clinker and/or MgO clinker. When the refractory material is used in a portion of a continuous casting apparatus having a heavy fusion damage in use, the MgO grain size should be adequately adjusted to prevent deterioration in quality of slabs due to outflow of the clinker into molten steel. Further, while a slight amount of Fe<sub>2</sub>O<sub>3</sub>, SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub> and/or ZrO<sub>2</sub> may be added to the clinker, it is desired to limit the amount of the additive within 10 mass%, because an excessive amount of the

additive is likely to cause deterioration in corrosion resistance and/or anti-alumina-buildup function.

While the  $\text{CaO} \cdot \text{MgO}$ -based refractory material can be applied to any refractory component for use in a continuous casting apparatus, such as a submerged nozzle, an upper nozzle, a lower nozzle, a sliding nozzle, a long nozzle, a stopper head, and a long stopper, it may be used in a given portion of such a refractory component to bring out an intended effect therein. In particular, the refractory material may be limitedly used in a portion of the refractory component to which a large amount of alumina is to be attached. For example, when the refractory material is used in one of the above nozzles, instead of making the entire nozzle of the refractory material, only the surface of the inner hole of the nozzle to be in contact with molten steel may be made of the refractory material to bring out an intended function sufficiently.

In the case of using the refractory material only in the surface of the inner hole of the nozzle, a green body of the composition according to the present invention may be placed on the inner surface and simultaneously shaped with the nozzle body or the composition according to the present invention may be inserted into the innerhole after shaped and baked into a sleeve and a ring, alternatively.

When the refractory material is used in the stopper head or long stopper, only the outer surface of the stopper head or long stopper to be in contact with molten steel may be made of the refractory material.

While the refractory material of the present invention is suitable for continuous casting of aluminum-killed steel, particularly aluminum-killed steel to be formed as a sheet metal, it may also be effectively applied to continuous casting of Al-Si killed steel, Al-Ti killed steel or Ti killed steel.

## BRIEF DESCRIPTION OF DRAWINGS

FIG. 1 is an electron micrograph showing the grain structure of  $\text{CaO} \cdot \text{MgO}$ -based clinker.

FIG. 2 is a graph showing the relationship between the mean diameter of MgO grains in the clinker and the mean diameter of MgO-based inclusions in slabs.

## BEST MODE FOR CARRYING OUT THE INVENTION

A preferred embodiment of the present invention will now be described in connection with examples.

### [EXAMPLE 1]

A plurality of samples different in mean diameter of MgO crystal grains were prepared by changing the cooling rate of electro-fused  $\text{CaO} \cdot \text{MgO}$ -based clinker having a chemical composition of 58 mass% of CaO and 41 mass% of MgO. Table 1 shows the respective mean diameters ( $\mu\text{m}$ ) of the prepared clinkers.

**Table 1**

MgO crystal	A	B	C	D	E	F	G	H	I
mean diameter ( $\mu\text{m}$ )	8	15	31	42	<50	61	69	78	88

The samples A to E are clinkers for Inventive Examples. For example, in the sample E, 60% or more of MgO crystal grains in the clinker have a diameter of 50  $\mu\text{m}$  or less. The samples F to I are clinkers for Comparative Examples. In the samples F to I, 60% or more of MgO crystal grains in each of the clinkers have a diameter of greater than 50  $\mu\text{m}$ .

Table 2 shows the compounding rate of each of compounds prepared by adding graphite and phenol resin to the respective clinkers in Table 1, and the frequency index of the occurrence of flaws during a rolling process in each of the compounds.

Table 2

		Inventive Example						Comparative Example			
		1	2	3	4	5		1	2	3	4
compounding rate mass%	code of CaO/MgO clinker sample	A	B	C	D	E		F	G	H	I
	1 - 0.5mm	30	30	30	30	30		30	30	30	30
	0.5mm >	15	15	15	15	15		15	15	15	15
	0.2mm >	30	30	30	30	30		30	30	30	30
	graphite 0.5 mm or less	25	25	25	25	25		25	25	25	25
	phenol resin	proper quantity	proper quantity	proper quantity	proper quantity	proper quantity		proper quantity	proper quantity	proper quantity	proper quantity
frequency of occurrence of flaws during rolling process (index) *		100	101	103	107	110		135	180	290	381

\* Inventive Example 1 = 100 (index having a smaller numeral indicates better quality of slabs)



Each of submerged nozzles serving as test pieces was prepared by forming a green body which has a powder line portion using a zirconia/graphite material and a nozzle body with an inner hole using each of the compounds in Table 2, through CIP forming at a forming pressure of 1000 kg/cm<sup>2</sup>, and burning the green body at a maximum temperature of 1000 °C in a reduction atmosphere.

Each of the submerged nozzles was used for continuous casting of aluminum-killed steel, and the quality of obtained slabs was checked. The casting was carried out under the conditions of a ladle volume: 250 ton, a TD volume: 45 ton, and a draw speed of slabs: 1.0 to 1.3 m/min.

The obtained slabs were milled to a sheet metal having a thickness of 2 mm, and the frequency of occurrence of flaws due to MgO-based inclusions was checked. The frequency of occurrence was indexed by defining the frequency of Inventive Example 1 as 100. The index having a smaller numeral indicates better quality of slabs. This test result shows that the quality of slabs is drastically improved in Inventive Examples where 60% or more of MgO crystal grains in the clinker have a diameter of 50 μm or less, as compared to Comparative Examples where the mean diameter is greater than 50 μm.

#### [EXAMPLE 2]

Table 3 shows the compounding rate of each of compounds prepared by uniformly kneading the respective CaO · MgO-based clinkers A to I in Table 1 and polypropylene added thereto, and the frequency index of the occurrence of flaws during a rolling process in each of the compounds. Each of upper nozzles serving as test pieces was prepared by forming a green body using each of the compounds in Table 3, through press forming at a forming pressure of 1200 kg/cm<sup>2</sup>, and burning the green body at a temperature of 1600 °C. Each of the upper nozzles was used for continuous casting of aluminum-killed steel under the same conditions as those in EXAMPLE 1, and the quality of obtained slabs was checked. This test result shows that the quality of slabs is drastically improved in Inventive Examples where 60% or more of MgO crystal grains in the clinker have a diameter of 50 μm or less.

Table 3

		Inventive Example					Comparative Example			
		6	7	8	9	10	5	6	7	8
compounding rate mass %	code of CaO/MgO clinker sample	A	B	C	D	E	F	G	H	I
	1 - 0.5mm	40	40	40	40	40	40	40	40	40
	0.5mm >	20	20	20	20	20	20	20	20	20
	0.2mm >	40	40	40	40	40	40	40	40	40
frequency of occurrence of flaws during rolling process (index) *	polypropylene	proper quantity	proper quantity	proper quantity	proper quantity	proper quantity	proper quantity	proper quantity	proper quantity	proper quantity
		100	102	105	109	112	129	173	321	453

\* Inventive Example 6 = 100 (index having a smaller numeral indicates better quality of slabs)

As mentioned above, the refractory material and the method for use in continuous casting of molten steel for sheet metal can reduce the mean diameter of MgO grains as inclusions in slabs obtained through the continuous casting. Thus, the frequency of occurrence of flaws in the milled sheet metal can be reduced to achieve a higher quality level of sheet metal products and the reduction of production cost.

### INDUSTRIAL APPLICABILITY

The refractory material and the method of using the refractory material of the present invention can be used in continuous casting of molten steel for sheet metal, particularly aluminum-killed steel for sheet metal.